Anomalous Heat Generation in Charging of Pd Powders with High Density Hydrogen Isotopes, (II) Discussions on Experimental Results and Underlying Physics

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Abstract: By our D(H)-gas charging experiments using Pd/PdO/ZrO2 powders, we obtained: 1) D-gas charge in the 1st phase (zero pressure interval) gave 20~90 % more excess heat than those for H-gas charge. 2) In the 2nd phase of pressure rise (finally up to 1MPa), significant excess heat (about 2 kJ/g-Pd) for D-gas charge was observed, while near zero level excess heat for H-gas charge was observed. We discuss the underlying surface and nano-particle physics in views of the enhanced surface adsorption potential by fractal sub-nano-scale trapping points on nano-Pd grain, the diffusion to inner shallower Bloch potential of regular Pd lattice, and the drastic mesoscopic and isotopic effect of surface and lattice rearrangement of nano-Pd particle by full D(H)-absorption to make deeper D(H) trapping potentials (about 2eV for D).

1. Introduction

The aim, experimental apparatus, experimental procedure and observed results of deuterium (and protium) gas charging experiments with various nano-fabricated Pd powders are written in our paper by Sasaki to this meeting. The observed results for Pd/PdO/ZrO2 nano-composite samples (about 10 nm diameter Pd particles dispersed in about 7 micron size ZrO2 flakes) provided us a confirmation-trial of Arata and Zhang experiment and a very interesting performance of deuterium (D) and protium (H) absorption and exothermic energy generation. The results by nano-Pd/ZrO2 samples are very interesting since specific surface effects in adsorption and following absorption into inner “lattice” sites look taking place to result in anomalously large stoichiometry values (x>1) of PdDx and deep trapping potential (or released energy). Compared with the results of 100nm Pd particle-powders, 10nm Pd particles dispersed in ZrO2 are regarded to have shown the drastic mesoscopic effects with isotopic difference.

2. Trend of Heat-Power Evolution

The evolutions of heat and gas-pressure can be discussed for two different phases, the 1st phase and the second phase as we show typical data in ‘Fig.1’.

The 1st phase is defined as the time-interval where reaction chamber (cell) keeps nominal “zero” gas-pressure. This means almost all D(H)-gas charged was absorbed by nano-Pd powders. Heat-power evolution curve in the 1st phase may be regarded mostly by chemical reaction energy during the D(H)-gas absorption into nano-Pd powders, in the ponit of view of conventional chemistry. However, we may have a component of “nuclear” process that we discuss later. For determining a 1st phase interval for heat-power evolution, we need to take a time-delay (about 15 minutes or three times of calorimetry time constant 5 minutes) into account, while the evolution of pressure has quick response. Trend of heat-power evolution in the second phase is very isotope dependent, as typical data is shown in ‘Fig.2’.

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Fig. 1. - Typical heat-power evolution data with Pd/ZrO₂ nano-composite sample with D-gas charge (D-PZ1#1 run).

Fig. 2. - Typical “excess heat-power” evolution by D-charge for (c) Pd/ZrO₂ sample, net Pd weight = 3g (D-PZ1#1 run), together with heat-power level by H-gas charge (H-PZ2#1 run), compared with (a) data by 100nm Pd powder, net Pd weight = 5g, and (b) data by Pd-black, net Pd weight = 3.2g (D) and 3.6g (H).

The H-gas charge has given “zero” power level (sometimes negative integrated values according to a possible zero-level drift of calorimetry or due to possible de-oxidation of PdO by D(H) charge). Obviously, the D-gas charge to the Pd/PdO/ZrO₂ nano-composite sample gave much more heat-power than that of the H-gas charge, for (c), which effect looks anomalously large compared to those by (a) and (b) and suggests a mesoscopic effect of Pd nano-particles on heat evolution, as discussed below.

3. Results and Discussions for 1st Phase Data

We summarized the integrated data of D(H)/Pd ratios, Heat per one-gram-Pd, Energy per D(H) atom absorption (E₁,₂), and gas-flow rates, and results are given in Table-1 of Reference-1 (see also Ref.8). First we discuss the data for 100nm Pd-particle powders (D-PP and H-PP runs in Table1 of Ref-1). Loading ratios, D(H)/Pd, are 0.43 and 0.44 respectively for deuterium (D) and protium (H) gas charging. Specific energies per absorbed D (or H) atom E₁ (or ΔH₁) values are 0.24eV/atom-D and 0.20eV/atom-H. When D(H) is absorbed in metal lattice, surface adsorption works on D(H)-molecule (or atom) first and D(H)-atoms diffuse into inner lattice sites (O-sites of Pd, usually) gradually. ‘Figure 3’ illustrates typical form of surface trapping potential and inner periodical (Bloch) trapping potentials.
For known values in text book\(^3\), \(E_{\text{th}}\) is about 0.5eV and \(E_{\text{hydr}}\) is 0.23eV. The difference, \(E_{\text{th}}-E_{\text{hydr}}\), is close to the energy released per D(H)-atom absorption in lattice and is about 0.25eV for bulk Pd lattice. Observed \(E_{\text{1st}}\) (\(\Delta H_s\)) values for 100nm Pd particle powder are near to this value of bulk Pd metal. This means that 100nm Pd particle works as bulk-metal for D(H) absorption. We will see \(E_{\text{1st}}\) values for Pd-black and Pd/PdO/ZrO\(_2\) nano-composite samples will have given much larger energies (deeper trapping potentials) to show the drastic mesoscopic effects. For the definition of mesoscopic size, we consider a Pd grain having 2,000 to 20,000 of total number of atoms.

Next, we look integrated data for Pd-black samples (D-PB and H-PB runs) in Table-1 of Ref.1. \(E_{\text{1st}}\) values in averages of runs are 0.70±0.15eV/atom-D and 0.69±0.10eV/atom-H. These values are significantly higher than the bulk value about 0.25eV. For the virgin runs (#1 runs), significantly high loading as PdD\(_{0.88}\) or PdH\(_{0.79}\) were observed. However, for runs with used samples (#2, #3, #4 runs), loading ratios were as small as 0.23 in average. Nevertheless the specific \(E_{\text{1st}}\) values were observed as same as the virgin (#1) runs. This fact means that microscopic active adsorption sites on surface of used Pd-black are working in the same way as the virgin Pd-black sample, although effective area of active surface decreased to 1/3 or less.

Now we discuss the integrated data in the 1\(^{st}\) phases for Pd/PdO/ZrO\(_2\) samples (D-PZ and H-PZ runs in Table1 of Ref.1). We observed heat-power levels were strongly dependent on the D(H) gas-flow rate. The larger gas-flow rate has trend to give the larger excess heat level in the 1\(^{st}\) phase, but the 1\(^{st}\) phase ends earlier than the case of smaller flow rate. This is understood as the faster gas-flow meets faster saturation of D(H) absorption in powders. Therefore, to compare specific values of \(E_{\text{1st}}\) (released energy per D(H)-atom), D(H)/Pd (loading ratio) and heat (in kJ) per g-Pd is more appropriate to understand the underlying physics.

Our first surprise is that all measured loading ratios, D(H)/Pd values for the 1\(^{st}\) phases are greater than 1.0, namely overloading (\(x=1.1\) in average for PdD(H)\(_x\) stoichiometry) in usual sense, even though the background gas pressure were nearly zero (near vacuum). This must be said as the drastic mesoscopic effect of D(H) absorption by the Pd nano-composite samples. For the known bulk Pd-metal, D(H) atoms are trapped in Bloch potentials (see Fig.3) at O-sites for \(x<1.0\). The observed anomalous data of \(x>1.0\) should show that the additional trapings at T-sites or surface happened by the mesoscopic effect (about 5,000 Pd atoms existing in a 5nm diameter particle).

![Fig. 3](image_url)

**Fig. 3.** Image of D(H) trapping potentials at surface adsorption (depth \(E_{\text{th}}\)) and lattice absorption (depth \(E_{\text{hydr}}\)); trapped D(H) atom diffuses gradually into inner Bloch trapping potentials through the QM tunneling. After full loading (\(x=1\) for PdD\(_x\)), rearrangement of Pd lattice may happen to make shallower potentials for bulk Pd (left figure), while nano-Pd grain in ZrO\(_2\) flake will have a specific mesoscopic potential (right figure) which reflects a non-linear combination of a deep collectively formed mesoscopic trapping potential (well-shape) and periodical Bloch potential of “local” PdD\(_x\) lattice.
The data for specific released energy $E_{1st}$ values are also anomalously large and isotope (D or H)-dependent. These are 2.2-2.5eV/atom-D and 1.3-2.1eV/atom-H. Deuterium gives larger $E_{1st}$ values. These released energy values are 5-10 times of the conventional value 0.25eV for bulk Pd metal. These values are however dependent on gas-flow rate, and we need further investigation so far. As PZ samples gave drastic mesoscopic effects, compared to the 100nm Pd powder we need further study by changing nano-Pd particle size.

In ‘Fig.4’ degassing data of Pd/ZrO$_2$ sample is shown compared with that of Pd-black, under baking procedures. Note that the vertical scale of pressure is logarithmic. Clearly, Pd/ZrO$_2$ sample can very highly retain D(H) atoms even after the evacuation. The desorption process of nano-Pd grain is speculated to be very unusual. The D(H) retention of Pd/ZrO$_2$ sample seems about 100-fold greater than that of Pd-black. This degassing results, showing peak at about 200ºC, also suggest that the D(H) trapping potential of Pd/ZrO$_2$ nano-composite sample is much deeper than that of bulk Pd lattice.

4. Excess Heat for the Second Phase

As summarized in Table-1 of Ref-1, we observed positive excess heat in the 2nd phase of D-PZ series runs for virgin (#1 run) samples (we tried three pairs of samples for simultaneous runs with D-gas by A1 (A2 for D-PZ5) and H-gas by A2 (A1 for H-PZ6) cells). For the D-PZ5 run, D-gas cylinder dried out in the 2nd phase and gas-pressure decreased (we had leakage), so that excess heat phenomenon was not observed. For earlier two PZ runs, we observed clear excess heat, only for D-gas charging, 2.27 and 2.07kJ/g-Pd in the time interval of 1,600 minutes, respectively.

Now, we discuss about a possible chemical energy release by oxidation$^8$ of charged (D(H)-gas, as samples contained PdO and ZrO$_2$ components. Already ‘Fig.4’ shows clearly D(H)-gas was mostly absorbed by Pd-grains in PZ samples, since Pd-black gave the same pattern of degassing. We have to consider a reduction of PdO$\_2$ states followed by production of $x$D$_2$O ($x$H$_2$O) and Pd$_x$ (PdH$_x$) states. The reaction energies $Q_l$ and $Q_H$ are evaluated to be $(162.6x + 70.0xy)$ kJ and $(156.6x + 58.0xy)$ kJ, respectively. For assumed values of $x = 1 \sim 0$ and $y = 0 \sim 1$, $Q_l$ and $Q_H$ are $0.84 \sim 0.73$ eV/D and $0.81 \sim 0.60$eV/H, respectively. These are too small to account for both the observed $E_{1st}$ energies and the isotope effects. Consequently, the main component of heat for a 1st phase cannot be due to the formation of water (D$_2$O or H$_2$O) by the D(H)-gas charging to PdO nano-particles. A role of PdO surface layer of Pd-nano-particles for making sub-nano-holes on surface under D(H)-charge will be discussed in our future paper.

There might be an yet-unknown atomic/electronic process governing the phenomenon in the present mesoscopic system, or the concept of ‘atom clusters’ might apply. However, it seems rather difficult to assume that such a large isotope effect is only in the electronic process of adsorption and/or hydride formation. Some nuclear process as suggested by the 4D/TSC model by Takahashi$^{4-7}$ could be a candidate mechanism responsible for the phenomenon.

A speculation for the underlying physics of the atomic & molecular-level cluster formation by Coulombic interaction of deuterons and electrons in metal-hydride and the strong nuclear interaction of multi-body deuteron interaction is illustrated in ‘Fig.5’, viewing the aspect from a few body system to more complicated interaction of deuterons and electrons in metal-hydride and the strong nuclear interaction of multi-body systems, namely, cluster, mesoscopic and bulk lattice systems. The 4D/TSC theory predicts a very rapid (in 1.4 fs) condensation$^{5-7}$ of 4D-cluster keeping the tetrahedral symmetric configuration with $P_d/ZrO_2$ powder in vacuum will be also explained by such a mesoscopic potential. Nuclear strong interaction for simultaneous 3 and 4 deuterons in very condensed 4D/TSC is modeled$^{4-7}$ based on the pion-exchange model (Hideki Yukawa) and potential (Hamada-Johnston). The pion exchange force for many body interaction has been scaled by the unit of PEF.
Fig. 4. - De-gassing data for Pd/ZrO$_2$(Santoku)-sample compared with that for Pd-black; Santoku-sample retains 100 times D(H)-atoms, presumably by its very deep trapping potentials of the mesoscopic effect of dispersed Pd-nano particles.

Fig. 5. - Speculation of potential change from a few body interaction to cluster, mesoscopic grain and bulk lattice of metal-deuterium (-protium) system, both for Coulombic and strong nuclear interaction. The proposed condensed cluster fusion is speculated to take place for the cluster, mesoscopic grain systems and near-surface of bulk metal as Pd.

5. Concluding Remarks

For Pd/PdO/ZrO$_2$ powders (Santoku-samples), we obtained results 1) through 4):

1) The D-gas charge in the 1st phase (zero pressure) gave 20-90% excess heat than the H-gas charge.
2) In the 2nd phase, significant excess heat (about 2 kJ/g-Pd) for the D-gas charge, while zero level for the H-gas charge, was observed.
3) No increase of neutron counts was seen, neither increase of gamma-rays over natural backgrounds.
4) D(H)/Pd ratios in the end of 1st phase was x>1.0, namely over-loading (x=1.1 in average). Flow rate dependence of x-values should be investigated further. Nano-Pd dispersed sample (Santoku, Pd/ZrO$_2$) retained 100 times more D(H) atoms after evacuation, than the Pd-black case.
We discussed that there may be a strong mesoscopic effect by Pd-nano-particle, namely surface and lattice rearrangement probably makes deep D(H) trapping potentials (1.0-2.5eV) by a nonlinearly coupled vibration states, of trapped D(H)-particles, between a deep collective trapping well potential and a periodical Bloch potential.

References


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